

Low-Temperature Synthesis of  $\text{NdFe}_{11}\text{TiC}_x\text{N}_y$  via Plasma Carbonitriding

Ken-ichi MACHIDA,\* Akira NAKAMOTO, and Gin-ya ADACHI\*

Department of Applied Chemistry, Faculty of Engineering,  
Osaka University, Yamadaoka, Suita, Osaka 565

Carbonitriding characteristics of  $\text{NdFe}_{11}\text{Ti}$  by applying the plasma technique to it in a  $\text{CH}_4\text{-N}_2\text{-H}_2$  mixed gas were studied. Carbon and nitrogen were simultaneously introduced to the  $\text{NdFe}_{11}\text{Ti}$  crystal lattice even at 423 K, of which the electronic states were found to be of metal carbide and nitride, respectively.

A series of rare earth intermetallic compounds containing nitrogen, viz.  $\text{Sm}_2\text{Fe}_{17}\text{N}_x$  and  $\text{NdFe}_{11}\text{TiN}_x$ , have been recently receiving much attention as new materials for high-performance permanent magnets, because they possess excellent magnetic properties, viz., high Curie temperature, strong uniaxial anisotropy, and high saturation magnetization.<sup>1)</sup> Furthermore, the compounds containing both carbon and nitrogen, e.g.  $\text{Sm}_2\text{Fe}_{17}\text{C}_x\text{N}_y$ , have been found to have the higher coercive force than the compounds containing only nitrogen, e.g.  $\text{Sm}_2\text{Fe}_{17}\text{N}_x$ .<sup>2,3)</sup>

Recently, we have applied a plasma technique to the nitriding of  $\text{Sm}_2\text{Fe}_{17}$  and succeeded in synthesizing  $\text{Sm}_2\text{Fe}_{17}\text{N}_x$  at the lower temperature than that for the conventional thermal reaction.<sup>4)</sup> In this work, the carbonitriding of  $\text{NdFe}_{11}\text{Ti}$  was carried out via the reactive species produced by glow-discharge in a  $\text{CH}_4\text{-N}_2\text{-H}_2$  mixed gas, and the resulting materials were characterized by means of X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS).

The intermetallic compound,  $\text{NdFe}_{11}\text{Ti}$ , was supplied by Santoku Kinzoku Inc., followed by annealing in Ar at 1373 K for 2 weeks and grinding up to a particle size below  $50\mu\text{m}$  in  $\text{N}_2$ . Plate samples (ca.  $5 \times 5 \times 2 \text{ mm}^3$ ) for XPS measurements were prepared by cutting the as-obtained (annealed) ingot, polishing with emery papers, and washing with acetone. The gases for plasma treatments,  $\text{CH}_4$  (99.9%) and  $\text{N}_2$  (99.99%), were purified by the method described elsewhere,<sup>4)</sup> but the  $\text{H}_2$  gas (99.999%) was not.

A series of plasma treatments were performed in the  $\text{CH}_4\text{-N}_2\text{-H}_2$  mixed

gas (molar ratio=1:1:4) at 423-723 K using the apparatus reported previously.<sup>4)</sup> The resulting materials were identified by XRD and XPS measurements, where the recorded XPS signals measured were calibrated on the basis of a C1s signal (273.0 eV). The nitrogen and carbon contents were also measured on them without any additional treatment, using Horiba EMGA-650 nitrogen and EMIA-521 carbon analyzers.

Figure 1 shows a series of XRD patterns of NdFe<sub>11</sub>Ti before and after the plasma treatment in the CH<sub>4</sub>-N<sub>2</sub>-H<sub>2</sub> mixed gas at 423-723 K. The composition and lattice parameters of the samples treated under conditions of 723 K and 1-4 h are summarized in Table 1. The XRD patterns of the treated samples were shifted to the low degree side of 2θ with the temperature and time for the plasma treatment, even at 423 K. In addition, the analysis data of composition demonstrated that the carbon and nitrogen are simultaneously introduced in the NdFe<sub>11</sub>Ti crystal lattice. It is, therefore, concluded that the plasma process can promote the carbonitriding reaction even at low temperature, while the effective thermal carbiding or nitriding starts only at the temperature above 700 K.<sup>2)</sup>

The crystallinity of samples was considerably decreased with the increase of the temperature for the plasma treatment. This may be due to the crystallographical limitation that the site for carbon or nitrogen in NdFe<sub>11</sub>Ti can accommodate only one atom per formula unit and hence the excess amount of them (x+y-1) is interstitially incorporated.

XPS signals of the N1s, C1s, and O1s electrons for the NdFe<sub>11</sub>Ti plate samples treated in the N<sub>2</sub>-H<sub>2</sub> and CH<sub>4</sub>-N<sub>2</sub>-H<sub>2</sub> mixed gases at 573 K for 2 h are shown in Figs. 2 (curves b and c), together with those of the as-obtained (untreated) sample (curves a). The C1s signals at ca. 285 and 289 eV were commonly observed on all the plate samples before Ar<sup>+</sup>-bombardment (see Fig. 2A), which were assigned to the carbon in

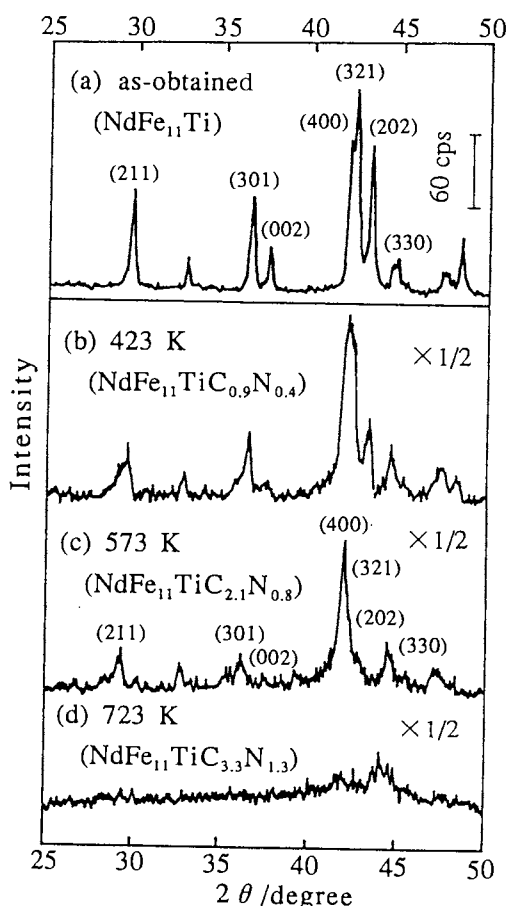


Fig.1. XRD patterns of as-obtained and plasma-treated NdFe<sub>11</sub>Ti powders. Conditions: in CH<sub>4</sub>-N<sub>2</sub>-H<sub>2</sub>, 423-723 K, 2 h.

Table 1. Composition and lattice parameters of the plasma-treated NdFe<sub>11</sub>Ti powder

| Treatment time/h <sup>a)</sup> | Composition  | Lattice parameters/Å |      |
|--------------------------------|--|----------------------|------|
|                                |  | a                    | c    |
| —                              | NdFe <sub>11</sub> Ti                                  | 8.57                 | 4.80 |
| 1                              | NdFe <sub>11</sub> TiC <sub>1.6</sub> N <sub>0.6</sub> | 8.65                 | 4.81 |
| 2                              | NdFe <sub>11</sub> TiC <sub>2.1</sub> N <sub>0.8</sub> | 8.66                 | 4.81 |
| 4                              | NdFe <sub>11</sub> TiC <sub>2.3</sub> N <sub>1.0</sub> | 8.66                 | 4.82 |

a) The treatment temperature was 573 K.

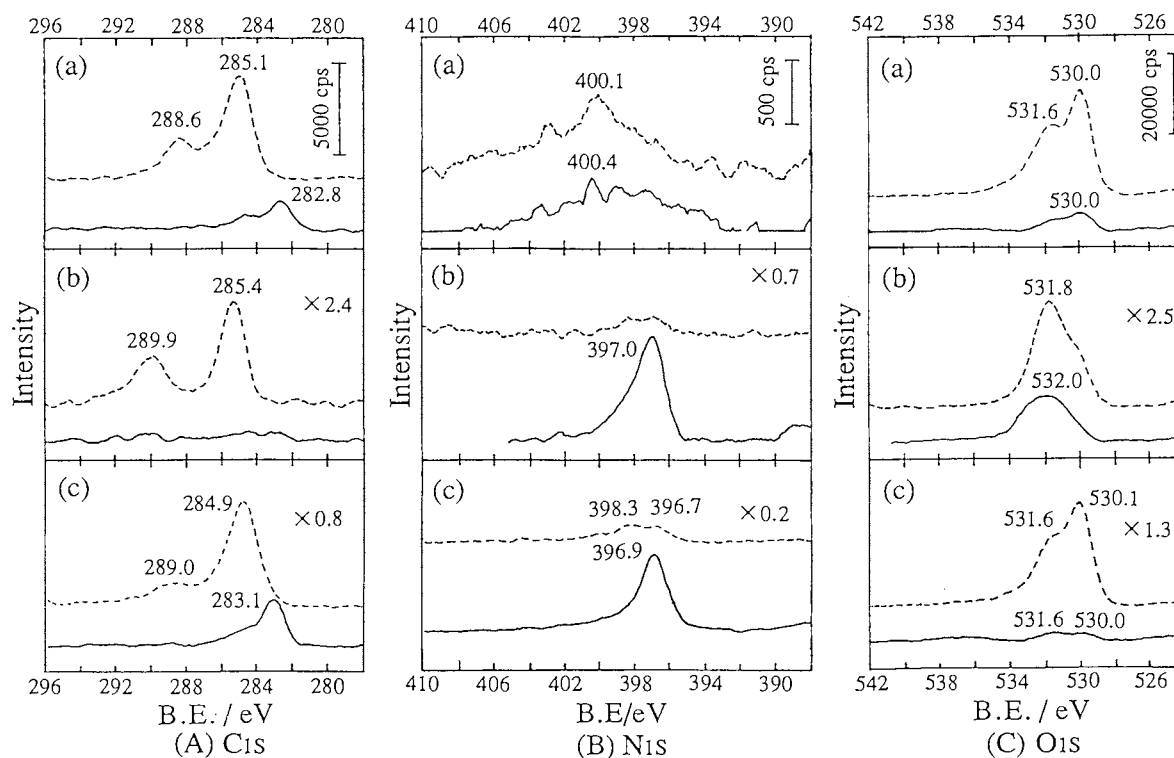


Fig.2. XPS signals of the C1s, N1s, and O1s electrons for the NdFe<sub>11</sub>Ti plates: (a) as-obtained, (b) plasma-treated in N<sub>2</sub>-H<sub>2</sub>, and (c) plasma-treated in CH<sub>4</sub>-N<sub>2</sub>-H<sub>2</sub>. Treatment condition: 573 K, 30 min. Sputtered depth: 0 Å for dashed lines and 250 Å for solid lines.

graphite and carbonate, respectively.<sup>5)</sup> However, these carbons are due to the surface contamination in the handling process of samples, because they are almost removed by Ar<sup>+</sup>-bombardment and, instead of them, new C1s signal assignable to the carbidic carbon appears at about 283 eV. It is, therefore, considered that the carbon incorporated in the NdFe<sub>11</sub>Ti crystal lattice form a metal carbide phase. The C1s signals observed on the

NdFe<sub>11</sub>Ti and NdFe<sub>11</sub>TiN<sub>x</sub> plates seems to be attributable to the carbon inherently contained in the as-obtained NdFe<sub>11</sub>Ti ingot.

For the N1s signals (see Fig. 2B), no apparent signal was observed on the as-obtained NdFe<sub>11</sub>Ti plate both before and after the Ar<sup>+</sup>-bombardment (curves a). However, the plasma-treated plates provided the N1s signal at about 396.7 eV even before the Ar<sup>+</sup>-bombardment (curves b and c), which was assigned to the nitrogen of metal nitrides.<sup>5)</sup> The fact that the intense N1s signal has been still observed even after the Ar<sup>+</sup>-bombardment indicates that the plasma nitriding effectively takes place up to the bulk level. An additional weak peak at 398.3 eV for the latter sample was assigned to cyanates.<sup>5)</sup>

In analogy with the Cls signals, the strong O1s signals were commonly observed at about 530-532 eV for all the samples. This indicates that the samples tend to be oxidized. However, the oxide formation is limited in the surface layer, particularly for the plasma-treated samples, since the signal intensity of O1s is decreased by the Ar<sup>+</sup>-bombardment.

In conclusion, the simultaneous introduction of carbon and nitrogen to the NdFe<sub>11</sub>Ti crystal lattice took place by the glow-discharge in the CH<sub>4</sub>-N<sub>2</sub>-H<sub>2</sub> mixed gas to form NdFe<sub>11</sub>TiC<sub>x</sub>N<sub>y</sub> via the simple one-step reaction. The XPS measurements demonstrated that the carbon and nitrogen in NdFe<sub>11</sub>TiC<sub>x</sub>N<sub>y</sub> exist as metal carbide and nitride, respectively.

This work was partially supported by a Grant-in-Aid for Scientific Research No.05650639 from the Ministry of Education, Science and Culture.

#### References

- 1) J. M. D. Coey and H. Sun, *J. Magn. Magn. Mater.*, **87**, L251 (1990); Y.-C. Yang, X.-D. Zhang, L.-S. Kong, Q. Pan, S.-L. Ge, J.-L. Yang, Y.-F. Ding, B.-S. Zhang, C.-T. Ye, and L. Jin, *Solid State Commun.*, **78**, 317 (1990); K. Kobayashi, T. Iriyama, N. Imaoka, and N. Kashiwaya, *European Patent Appl. Publ.*, No. 0-417-732-42 (1990).
- 2) "Handbook of Magnetic Materials," ed by K. H. J. Buschow, Vol. 6, North-Holland, Tokyo (1991).
- 3) X.-C. Kou, R. Grossinger, M. Katter, J. Wecker, L. Schultz, and K. H. J. Buschow, *J. Appl. Phys.*, **70**, 2272 (1991).
- 4) K. Machida, E. Yamamoto, K. Murase, G. Adachi, M. Taniguchi, and K. Tanaka, *Chem. Lett.*, **1992**, 1243.
- 5) C. D. Wagner, W. M. Riggs, L. E. Davis, J. F. Moulder, G. E. Muilenberg, "Handbook of X-ray Photoelectron Spectroscopy," Perkin Elmer Co., Physical Electronics Div. (1979).

(Received May 7, 1993)